## Synthesis and Photoelectrochemical Behavior of Nitrogen-doped NaTaO<sub>3</sub>

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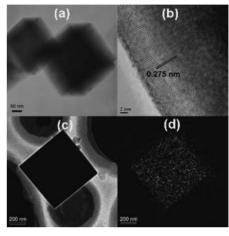
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Well-dispersed NaTaO<sub>3</sub> particles with a cubic morphology and high crystallinity were prepared by a hydrothermal route. The NaTaO<sub>3</sub> was doped with nitrogen by heat treatment in ammonia atmosphere. The nitrogen-doped NaTaO<sub>3</sub> shows an additional visible light absorption shoulder up to 550 nm. The photoelectrochemical investigation demonstrated that the photocurrent density of the photoanode of the nitrogen-doped NaTaO<sub>3</sub> was enhanced under the irradiation of both UV–visible light and visible light, compared to that of the undoped NaTaO<sub>3</sub>.

Recently, more and more efforts<sup>1-3</sup> have been devoted to developing visible-light-responsive photocatalysts. Among the many photocatalysts discovered to date, NaTaO3 shows high photoactivity. However, the band gap of NaTaO3 is ca. 4.0 eV and does not exhibit visible light absorption. Doping is considered to be an effective route to improve the photocatalytic performance of photocatalysts. 4,5 Nitrogen doping, especially in titania-based photocatalysts, <sup>6,7</sup> has been extensively investigated, and many routes have been explored.<sup>8-11</sup> Among these routes, the most simple and facile one is thermal treatment in ammonia atmosphere, especially for multicomponent compounds derived via traditional solid-state reactions. The advantage of this route lies in that heating can supply enough energy for incorporating N atoms, which is essential for the realization of nitrogen doping in multicomponent compounds with a large crystal size and high crystallinity. So far, nitrogen doping has been applied to many multicomponent systems, such as ATaO<sub>2</sub>N (A = Ba, Sr, and Ca) and LaTaON<sub>2</sub>. <sup>12,13</sup> However, nitrogen doping has rarely been realized in NaTaO3 systems, which is the most efficient for photoelectrochemical splitting of pure water to hydrogen under UV irradiation, 14 because NaTaO<sub>3</sub> obtained from solid-state reactions is easily transformed to Ta<sub>3</sub>N<sub>5</sub>.

In this study, we report nitrogen-doped  $NaTaO_3$  by heat treatment of hydrothermally synthesized cubic  $NaTaO_3$  in ammonia atmosphere. After nitrogen doping, the crystal structure of  $NaTaO_3$  is well retained, but the band gap of  $NaTaO_3$  decreases from ca. 4.0 to 3.7 eV. Furthermore, an additional visible light absorption in the photon energy up to ca. 2.25 eV is introduced. Compared to the undoped  $NaTaO_3$ , the nitrogen-doped  $NaTaO_3$  photoanode is more active under the irradiation of visible light in photoelectrochemical processes.

For NaTaO $_3$  preparation, 0.2 mol of TaCl $_5$  (99.99%, Alfa Aesar) was dissolved in 10 mL of 7 M HCl solution. Diluted ammonia was slowly added dropwise into the hydrochloric acid solution of TaCl $_5$  to neutralize it for the hydrolysis of TaCl $_5$  (pH 7). The precipitate obtained was added to 60 mL of 2 M NaOH solution to obtain a suspension, and then the suspension was placed into a Teflon autoclave and heated at 140 °C for 20 h.



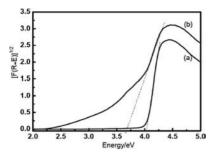
**Figure 1.** TEM image (a), HRTEM image (b), energy-filtered TEM image (c), and N mapping (d), of the nitrogen-doped NaTaO<sub>3</sub>.

After washing and drying at  $60\,^{\circ}\text{C}$  for  $12\,\text{h}$ , the undoped NaTaO<sub>3</sub> was obtained. Nitrogen-doped NaTaO<sub>3</sub> was prepared by heating the NaTaO<sub>3</sub> under NH<sub>3</sub> flow of  $25\,\text{mL}\cdot\text{min}^{-1}$  at  $850\,^{\circ}\text{C}$  for  $4\,\text{h}$ .

Compared to the irregular NaTaO $_3$  particles derived from traditional solid-state reaction routes, the NaTaO $_3$  particles we prepared are cubic-shaped with a single crystal structure. After nitrogen doping, XRD patterns of the nitrogen-doped NaTaO $_3$  remain the same as the undoped NaTaO $_3$  (Figure S1), $^{22}$  and both crystal structure (Figure S1) $^{22}$  and particle morphology (Figure 1a and Figure S2) $^{22}$  of the undoped NaTaO $_3$  are well retained. The lattice spacing of 0.275 nm in Figure 1b is consistent with the lattice spacing of the {200} planes of cubic NaTaO $_3$ . In contrast to the reported hydrothermal route  $^{15,16}$  with Ta $_2$ O $_5$  as precursor, our route has the merit of low concentration of NaOH solution, high dispersion, and pure single phase, which are helpful for nitrogen doping.

Energy-filtered TEM was used to determine the elemental mapping of N in the photocatalyst. Figure 1c shows the general morphology, and Figure 1d is N mapping of the nitrogen-doped NaTaO<sub>3</sub>. These results show that nitrogen atoms are homogeneously distributed in the nitrogen-doped NaTaO<sub>3</sub> particles. XPS results (Figure S3)<sup>22</sup> confirm the existence of the oxidation state of N (395.2 eV) in nitrogen-doped NaTaO<sub>3</sub>, and the content of N dopant on the surface of the nitrogen-doped NaTaO<sub>3</sub> is estimated to be 1.5 atom %.

Compared to the undoped NaTaO<sub>3</sub>, the band gap of nitrogen-doped NaTaO<sub>3</sub> decreases from ca. 4.0 to 3.7 eV (Figure 2). In addition, the absorption edge of NaTaO<sub>3</sub> after nitrogen doping is extended to ca. 2.25 eV, which means light absorption extend-

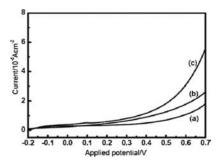


**Figure 2.** Plots of transformed Kubelka–Munk function vs. the energy of light absorbed of (a)  $NaTaO_3$  and (b) nitrogen-doped  $NaTaO_3$ .

ing to ca. 550 nm. For metal oxide semiconductors, 17,18 the valence bands consist of main O 2p states and partial metal d states. The mechanism for introducing visible light absorption in anion-doped metal oxide semiconductors is either generating localized states of anion dopants or narrowing the band gap by elevating the valence band maximum. In our nitrogen-doped NaTaO<sub>3</sub>, both mechanisms are involved, and O-Ta-N bonds and N-Ta-N bonds exist. The mixture of N 2p with O 2p states caused by the stronger interaction between Ta and N in N-Ta-N bonds results in the decreased band gap of the nitrogen-doped NaTaO<sub>3</sub>. Compared to that of N atoms in N-Ta-N bonds, the binding energy of N atoms in O-Ta-N bonds is lower owing to stronger electronegativity of O than N. So the weaker interaction between Ta and N in O-Ta-N bonds gives rise to the localized states with higher energy level in the band gap, which is responsible for the additional shoulder of light absorption in the nitrogen-doped NaTaO<sub>3</sub>. Moreover, oxygen vacancies, inevitably existing in the nitrogen-doped NaTaO<sub>3</sub> system in order to keep the charge balance, give rise to the localized states with higher energy level in the band gap, which is also responsible for the additional shoulder of light absorption.

Photoelectrochemical measurements are widely employed to determine the photoactivity of semiconductor photoanodes. 19-21 Following reported procedures, 20 the ITO photoanode consisting of the nitrogen-doped NaTaO3 shows higher photocurrent than that with NaTaO3 under the irradiation of both UV-visible light (Figure S4)<sup>22</sup> and visible light (Figure 3). It is noted that the enhanced photocurrent is positively dependent on the increase of N dopant in NaTaO3 before the crystal structure is changed. The turn-on potentials of photocurrent for the anodes of the undoped and nitrogen-doped NaTaO3 under the irradiation of visible light are ca. 0.1 and -0.12 V. Generally, the turn-on potential can roughly determine the flat-band potential of the photoanode in a given electrolyte, while the flat-band potential is closely related to the Fermi level of the semiconductor photoanode. For nitrogen-doped NaTaO<sub>3</sub>, the introduced localized states in the band gap and mixed states of N 2p with valence band can exert a great influence on its Fermi level and hence its flat-band potential and turn-on potential. In this case, oxygen vacancies also contribute to the elevation of Fermi level. Obviously, the enhanced photocurrent for the nitrogen-doped NaTaO<sub>3</sub> photoanode can be attributed to the excited electrons from N 2p and oxygen vacancies local states.

In summary, we prepared well-dispersed and highly crystalline cubic  $NaTaO_3$  particles and then doped nitrogen into the  $NaTaO_3$  particles by thermal treatment in ammonia atmosphere.



**Figure 3.** Photocurrent–applied potential curves of NaTaO<sub>3</sub> photoanodes on ITO substrate vs. Hg/HgCl reference electrode: (a) photoanode with the nitrogen-doped NaTaO<sub>3</sub> in dark, (b) photoanode with the undoped NaTaO<sub>3</sub>, and (c) photoanode with the nitrogen-doped NaTaO<sub>3</sub> under the irradiation of visible light. Measurement condition: 0.1 M NaOH electrolyte solution, scan rate  $5 \,\mathrm{mV} \cdot \mathrm{s}^{-1}$ . The illuminated photoanode surface was  $1 \,\mathrm{cm}^2$ .

The prepared nitrogen-doped NaTaO<sub>3</sub>, compared to the undoped NaTaO<sub>3</sub>, exhibits increased light absorption from 300 nm up to 550 nm. Correspondingly, the photocurrent density of the nitrogen-doped NaTaO<sub>3</sub> photoanode is enhanced under the irradiation of both UV–visible light and visible light.

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